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W-8000 München 40(DE)(54) **Amorphous elastomeric propylene homopolymers.**

(57) An elastomeric high molecular weight substantially amorphous propylene homopolymer having a melting point between about 145°C and 165°C, a melt viscosity at 190°C greater than 200,000 cps and a heat of fusion between about 4 cal./gm. and about 10 cal./gm. is disclosed. The polymer comprises from about 35% to about 55% of a diethyl ether soluble fraction, the fraction having an inherent viscosity less than about 1.0 dl/g and exhibiting substantially no isotactic crystallinity and from about 30% to about 70% of a heptane soluble fraction, the fraction having a melting point between about 100°C and 115°C and a heat of fusion less than about 2.0 cal./gm.

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This invention relates to an elastomeric substantially amorphous propylene homopolymer. The polymer has properties which make it applicable for use as a thermoplastic elastomer in a variety of applications including films and film compositions.

Crystalline and amorphous polypropylenes are materials that are used in a wide variety of applications and billions of pounds of polypropylene homopolymers and copolymers are produced on an annual basis. Amorphous polypropylene is generally regarded as being predominantly comprised of an atactic propylene polymer. Alternatively, crystalline polypropylene is considered to be polypropylene that is comprised prevailing of syndiotactic or isotactic polypropylene. The difference between isotactic, syndiotactic and atactic polypropylenes lies in the stereochemistries of the polymers. The substituent methyl groups of isotactic polypropylene are located on one side of the extended chain that makes up the polymer whereas syndiotactic propylene has methyl groups that alternate regularly from side to side. The regularity of the isotactic and syndiotactic structures allows the molecules to fit together well, resulting in a highly crystalline material. The methyl groups of atactic polypropylene, on the other hand, are distributed randomly along the polymer chain. Atactic polypropylenes are generally gummy materials of little strength that are used in the production of adhesives. The structure of isotactic, syndiotactic and atactic polypropylene is illustrated on p. 1248 of Organic Chemistry by Morrison and Boyd, fifth edition.

Highly crystalline isotactic polypropylenes make up the majority of commercially available polypropylenes. The isotactic structure results in polypropylene with high stiffness and good tensile strength. Crystalline polypropylene, however, generally has a high tensile set and is not suitable for those applications where elastomeric properties are desirable.

In the past, the manufacture of propylene homopolymers and copolymers using conventional polymerization techniques and unsupported catalysts resulted in the simultaneous production of substantial quantities of atactic polymer in addition to the desired highly crystalline and predominantly isotactic product. Various methods were used to separate and purify the two products and the atactic by-product was used commercially as a component in adhesive compounds, roofing materials, caulking compounds and the like. Over the last decade, however, major advances have been made in catalysts for the production of isotactic polypropylene. The use of highly active stereospecific catalysts has resulted in the production of isotactic homopolymers and copolymers that do not require purification or removal of the atactic or low crystalline polymer. The development of novel high molecular weight atactic propylene polymers has, however, received much less attention. A fractionable elastomeric polypropylene having a high molecular weight diethyl ether-soluble fraction that has about 0.5% to about 5% isotactic crystallinity and an inherent viscosity exceeding 1.5 is disclosed in U.S. Pat. No. 4,335,225 to Collette et al., issued June 15, 1982. This polymer does not, however, have the properties of the novel polymer of the present invention.

The present invention provides a novel high molecular weight amorphous propylene homopolymer that has elastic properties. A diethyl ether soluble fraction of the polymer has no isotactic crystallinity and an inherent viscosity less than about 1.0 dl/g. The polymer is produced with a solid supported catalyst component and an organoaluminum component. The polymer of the present invention has a molecular weight such that the melt viscosity of the polymer is greater than 200,000 cps at 190°C and the melt flow rate (MFR) at 230°C is less than 80 g/10 minutes. The polymer is a general purpose thermoplastic elastomer adaptable to a wide variety applications including films, filaments, fibers, sheet, molded articles, and other applications where elastomeric properties are desirable. The polymer may also be blended with other homopolymers and copolymers to produce blends having useful properties.

In accordance with the present invention there is provided a substantially amorphous high molecular weight propylene homopolymer. The terminology "high molecular weight" as used herein refers to polymers having a melt viscosity at 190°C greater than 200,000 cps or a melt flow rate (MFR) at 230°C less than 80 g/10 minutes. The polymer includes a diethyl ether soluble fraction that exhibits no isotactic crystallinity and an inherent viscosity less than about 1.0 dl/g.

Isotactic crystallinity, as referred to herein, is determined by ¹³C Nuclear Magnetic Resonance (NMR). The ¹³C spectra are obtained at 125°C using a JEOL FX270 NMR spectrometer operating at a ¹³C frequency of 67.8 MHz with a proton decoupling of 269.65 MHz. A 12 microsecond pulse width (approximately 65°) and a 2.1 second pulse repetition rate is used for quantification. The samples are run as a solution of 0.1 gm of polymer in 2 ml of a 90/10 by weight solution of 1,2,4-trichlorobenzene and D₆-benzene. N-phenyl-1-naphthylamine is used as an antioxidant (approximately 0.04M).

When a sample of polypropylene is tested for isotactic content as set forth above and then cooled to ambient temperature and held for 24 hours there may be a reduction in the area under the mmmm pentad peak if there is crystallization of sufficiently long isotactic sequences. The decrease in mmmm intensity is attributed to immobilization of isotactic pentads by crystallization and thereby provides a measure of the fraction of polymer crystallized. An NMR rigidity parameter is defined by Collette et al., MACRO-

MOLECULES 22, 1358 (1989) as:

$$R_{nmr} = F_h - F_l(1 - F_h)/(1 - F_l)$$

- 5 Where F_h and F_l are the mmmm pentad fractions at the higher and lower temperatures, respectively. The relationship between

R_{nmr} and crystallinity is given as:

$$R_{nmr} = 0.006 + 0.016(\% \text{ crystallinity})$$

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An R_{nmr} of less than or equal to zero would, therefore, indicate the absence of isotactic crystallinity.

- The inherent viscosity of the polymer is determined by placing a 0.05 gm sample in 100 ml. of decahydronaphthalene containing 0.1 g/l BHT (2,6-di-t-butyl-4-methyphenol) and heating the sample to 135°C under nitrogen as it is stirred with a magnetic stirrer for two hours. The mixture is poured through a
15 filter stick into a Cannon-Ubbelohde viscometer and its efflux time is measured at 135°C. and compared with that of the solvent alone. The inherent viscosity is then calculated as:

$$\text{Inherent viscosity} = (\ln T/T_0)/C$$

- 20 Where T is the efflux time of the mixture, T_0 is the efflux time of the solvent and C is the concentration of the polymer in grams/deciliter.

- The solubility of the homopolymer in the solvents referred to herein is determined by placing a sample (6 grams) into a cellulose thimble. The thimble is placed into a vertical cylindrical chamber in a round bottom glass flask containing the solvent. The solvent is heated to boiling and solvent vapors rise through
25 an outer zone surrounding the cylindrical chamber and are cooled in a reflux condenser. The condensed solvent drips onto the thimble and extraction continues for 16 hours. The portion of the sample extracted is the soluble portion.

- Melting points and heats of fusion are measured with a Perkin Elmer DSC-2C Differential Scanning Calorimeter (DSC). Melting point is determined according to ASTM method D-3481. Melt flow rates are
30 measured using a Tinius-Olsen extrusion plastometer in accordance with ASTM method D-1238. The propylene homopolymer of the present invention has a melting point between about 145°C and about 165°C and a heat of fusion of from about 4 cal/gm to about 10 cal/gm. The polymer has a melt viscosity of from about 200,000 cps to greater than about 2,000,000 cps at 190°C and a melt flow rate of from about 4 gm/10 minutes to about 80 gm/10 minutes at 230°C.

- 35 The propylene homopolymer of the present invention comprises from about 35% to about 55% of a diethyl ether soluble fraction. The diethyl ether soluble fraction exhibits no ^{13}C isotactic crystallinity and has an inherent viscosity less than about 1.0 dl/g. The ether soluble fraction has no detectable melting point above 40°C., a further indication of the absence of crystallinity.

- The propylene homopolymer of the present invention comprises from about 30% to about 70% of a
40 heptane soluble fraction. The heptane soluble fraction of the homopolymer has a melting point between about 100°C. and about 110°C. and a heat of fusion less than about 2.0 cal/gm. The heptane insoluble fraction of the homopolymer has a melting point between about 145°C. and 165°C. and a heat of fusion between about 10 cal./gm. and about 20 cal/gm.

- The homopolymer of the present invention is prepared by a process which comprises polymerizing
45 propylene monomer at a temperature between about 120°F and about 175°F in the presence of a particular catalyst composition. Polymerization should be conducted at a pressure sufficient to maintain propylene in the liquid phase, usually pressures in the range between about 400 psig and about 550 psig are suitable. The preferred temperature is between about 130°F and about 150°F.

- Close control of the hydrogen concentration at a level such that the hydrogen partial pressure does not
50 exceed about 4.0% of the reactor pressure during the preparation of the homopolymer of the present invention is important to produce a polymer with a melt viscosity at 190°C greater than 200,000 cps. Preferably, no hydrogen is introduced during polymerization, producing a polymer with a melt viscosity at 190°C greater than 2,000,000 cps.

- The catalyst composition used to polymerize the amorphous high molecular propylene homopolymer
55 contains a solid, supported catalyst component and an organoaluminum component. The supported catalyst component is comprised of an active transition metal compound such as titanium tetrahalide mixed with an enhanced support comprised of magnesium halide and aluminum trihalide. The molar ratio of magnesium halide to aluminum trihalide is about 8:0.5-3.0 and preferably about 8:1.0-1.5.

The molar ratio of magnesium halide to titanium tetrahalide is between about 8:0.1-1.0 and preferably about 8:0.4-0.6. The preferred magnesium halide is magnesium chloride. An important step in the preparation of the solid supported catalyst component is the exclusion of electron donor compounds. Additionally, polymerization of the homopolymer with the catalyst should be carried out in the absence of added electron donors.

The catalyst system used to prepare the homopolymer is a composition of:

(A) a solid catalyst component produced by the method comprising:

(i) co-comminuting magnesium halide support base and aluminum trihalide in a molar ratio from about 8:0.5 to about 8:3 in the absence of added electron donor; and

(ii) then co-comminuting the product of step (i) in the absence of added electron donor with sufficient titanium tetrahalide to provide a molar ratio of magnesium halide to titanium tetrahalide from about 8:0.1 to about 8:1; and

(B) a trialkylaluminum co-catalyst component having from 1 to 9 carbon atoms in each alkyl group in a sufficient quantity to provide an Al/Ti ratio in the range from about 50:1 to about 600:1.

Thus, the modified method involves co-comminuting magnesium halide and aluminum trihalide in the absence of an electron donor and then co-comminuting the catalyst support so formed with titanium tetrahalide, also in the absence of an electron donor.

The solid catalyst component is used in conjunction with an organoaluminum co-catalyst, which is a trialkylaluminum wherein each alkyl group contains between 1 and 9 carbon atoms. Preferably, the alkyl groups are ethyl groups. The invention will be described hereinafter in connection with the preferred catalyst system. The molar ratio of organoaluminum co-catalyst to titanium-containing catalyst component, i.e., Al/Ti ratio should range between about 50:1 and about 600:1, preferably between about 90:1 and about 300:1.

The polymerization is carried out in a stirred reactor at average residence times between about 1 hour and about 3 hours. Sufficient catalyst quantities are fed to the reactor to result in a polymer solids concentration in the reactor slurry of from about 10 wt % to about 50 wt %. The reactor effluent is withdrawn from the reactor, and unreacted monomer and hydrogen is flashed from the polymer.

Various additives can be incorporated into the polymer, such as antioxidants, U.V. stabilizers, pigments, etc.

An additional benefit of the polymer of this invention is that since the polymer contains small quantities of catalyst residues due to the high productivity rates of the specific catalyst used in the polymerization, there is no need to remove these small amounts of catalysts from the polymer.

The invention will be further described with respect to the following examples; however, the scope of the invention is not to be limited thereby.

Example 1

The experiment was performed in a 1-liter, jacketed autoclave equipped with a magnetically coupled stirrer. The temperature of the autoclave was controlled using a mixture of equal weights of glycol and water as the heat transfer fluid flowing the jacket of the autoclave. The temperature of the fluid was controlled with a microprocessor connected to an iron/constantin thermocouple inside the autoclave. Set point temperature was maintained at $\pm 0.2^\circ\text{C}$. Propylene monomer was polymerization grade passed through molecular sieve beds, as well as beds of copper catalyst for oxygen removal prior to use. Triethylaluminum (TEA) was purchased as a 25% W/W in normal heptane and used as is. A 1% W/W slurry of the solid catalyst component was prepared using degassed mineral oil. Prior to use, the autoclave was heated to 90°C . with a slow nitrogen purge for 30 minutes, cooled to 30°C ., and purged with propylene vapor. The alkyl solution and catalyst slurry were prepared in septum vials in dry boxes and introduced into the reactor using hypodermic syringes.

1.98 ml. of TEA (25%) and 1.71 ml. of 1% catalyst slurry (2.5% W/W titanium content) were added to the reactor. 0.6 liters of propylene monomer (300 gm.) was introduced using a sight gauge and nitrogen pressure. The reactor content was heated to 60°C . and maintained for two hours while stirring at 500 rpm. After two hours, the temperature was rapidly lowered and the unreacted propylene vented. The polymer was dried under vacuum at 40°C . overnight. Test results are set forth in Table 1 below.

Example 2

Polymerization of propylene homopolymer was conducted in a large scale continuous pilot plant operation. Propylene monomer and catalyst components were separately and continuously charged to a

stirred reactor, the monomer feed rate corresponding to about a two hour residence time in the reactor. The organoaluminum compound of the catalyst system was a 5% W/W heptane solution of triethylaluminum (TEA). The solid supported titanium tetrachloride catalyst had a titanium content of about 2.5% W/W and was pumped into the reactor as a 6% W/W mixture in petrolatum. The two catalyst components were added at rates directly proportional to the polymer production rate and in amounts sufficient to maintain the polymer solids concentration in the reactor slurry in a range from about 10% to about 15%. The reaction temperature was maintained throughout at about 140°F. (60°C.). The catalyst efficiency was calculated from the polymer slurry withdrawal rate, solids content in the slurry, and the titanium catalyst addition rate. The polymer was separated from the unreacted monomer, stabilized with Irganox® 1010, pelletized, and tested. Test results are set forth in Table 1 below.

Table 1

	<u>Example 1</u>	<u>Example 2</u>
Melting point °C.	154.9	153.9
Heat of fusion cal./gm.	7.7	6.1
Melt flow rate gm./10 min. @ 230°C.	5.7	9.7
Melt viscosity cps @190°C.	>2,000,000	>2,000,000
Ether soluble fraction weight%	42.8	44.0
Inherent viscosity (ether soluble fraction)	0.93	0.82
Catalyst efficiency Kg polymer/g Cat.	11.8	6.8
Heptane soluble fraction weight %	43.8	62.8
Melting point °C. (heptane soluble fraction)	105.6	105.8
Heat of fusion cal./gm. (heptane soluble fraction)	0.6	1.6
Melting point °C. (heptane insoluble fraction)	156.1	151.1
Heat of fusion cal./gm. (heptane insoluble fraction)	14.4	15.1

Test bars of 0.125" thickness were produced from the polymer using a 75 Ton Van Dorn Injection Molding machine. These bars were tested for tensile strength at break, elongation at break, and tensile set. The tensile set of the polymer was 69% at 300% elongation. No break was observed at 700% elongation.

Ether soluble fractions of the polymers of Examples 1 and 2 were extracted and tested for NMR isotactic crystallinity as described above. The results are set forth below.

Table 2

	Example 1	Example 2
NMR isotactic content @ 125° C. (F _H)	0.16	0.15
NMR isotactic content @ 32° C. (F _I)	0.16	0.16
NMR rigidity	0	<0
NMR crystallinity	0	0

Examples 3 and 4

Propylene monomer was polymerized in the same fashion as in Example 1 except that hydrogen was added to the reactor at a partial pressure of 5 psig in Example 3 and 15 psig in Example 4. Test results are set forth in Table 3 below.

Table 3

	<u>Example 3</u>	<u>Example 4</u>
Melting point °C.	155.8	157.4
Heat of fusion cal./gm.	5.1	5.8
Melt flow rate gm./10 min. @ 230°C.	11.3	79.0
Melt viscosity cps @190°C.	>2,000,000	209,000
Catalyst efficiency Kg polymer/g Cat	12.5	13.9
Heptane soluble fraction weight %	56.4	60.9
Melting Pt. °C (heptane soluble fraction)	106.6	110.4
Heat of fusion cal./gm. (heptane soluble fraction)	1.0	1.1
Melting Pt. °C (heptane insoluble fraction)	157.7	158.4
Heat of fusion cal./gm. (heptane insoluble fraction)	17.0	17.6

Comparative Examples 5 and 6

In order to illustrate the importance of hydrogen control in the preparation of the homopolymer of the present invention, propylene monomer was polymerized in the same manner as in Example 1, except that hydrogen was added at partial pressures of 45 and 70 psig respectively. The polymers were tested and the results are set forth in Table 4 below.

Table 4

	<u>Example 5</u>	<u>Example 6</u>
H ₂ partial pressure (psig)	45	70
Melt flow rate g/10 min. @ 230°C	fluid	fluid
Melt viscosity cps @ 190°C.	6,300	1,730
Heptane soluble fraction wt. %	79.7	84.3
Melting pt. °C (heptane soluble fraction)	115.5	134.8
Heat of Fusion cal./gm. (heptane soluble fraction)	2.3	3.1
Melting pt. °C (heptane insoluble fraction)	151.4	152.2
Heat of Fusion cal./gm. (heptane insoluble fraction)	21.3	21.5

Comparative Examples 5 and 6 illustrate that the addition of excessive amounts of hydrogen results in the production of polymers having undesirably high melt flow rates and undesirably low melt viscosities.

While particular embodiments of the present invention have been described, it will be apparent to those skilled in the art that various changes and modification can be made without departing from the spirit and scope of the invention. The following claims are intended to cover all such modifications that are within the scope of the invention.

Claims

1. An elastomeric high molecular weight substantially amorphous propylene homopolymer having a melting point between about 145°C and 165°C, a melt viscosity at 190°C greater than 200,000 cps, a heat of fusion between about 4 cal./gm. and about 10 cal./gm., the polymer comprising from about 35% to about 55% of a diethyl ether soluble fraction, the fraction having an inherent viscosity less than about 1.0 dl/g and being substantially free of isotactic crystallinity.
2. The polymer of claim 1 wherein the homopolymer comprises from about 30% to about 70% of a heptane soluble fraction, the fraction having a melting point between about 100°C and 115°C and a heat of fusion less than about 2.0 cal./gm.
3. The homopolymer of at least one of the claims 1-2 wherein the homopolymer is polymerized with a catalyst system comprising:
 - (A) a solid catalyst component prepared by:
 - (i) co-comminuting magnesium halide support base and aluminum trihalide in a molar ratio from about 8:0.5 to about 8:3 in the absence of added electron donor; and
 - (ii) then co-comminuting the product of step (i) in the absence of added electron donor with sufficient titanium tetrahalide to provide a molar ratio of magnesium halide to titanium tetrahalide from about 8:0.1 to about 8:1; and
 - (B) a trialkylaluminum co-catalyst component having from 1 to 9 carbon atoms in each alkyl group in

a sufficient quantity to provide an Al/Ti ratio in the range from about 50:1 to about 600:1.

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Application Number

DOCUMENTS CONSIDERED TO BE RELEVANT			EP 91115127.2
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl.5)
A	<u>DE - A - 3 246 479</u> (CHISSO CORP.) * Claims 1-2; examples * --	1, 2	C 08 F 110/06 C 08 F 4/646
A	<u>EP - A - 0 114 391</u> (SUMITOMO CHEMICAL COMPANY, LIMITED) * Claim 1; examples * --	1, 2	
D, A	<u>US - A - 4 335 225</u> (COLETTE et al.) * Claim 1 * --	1	
A	<u>US - A - 4 847 340</u> (ALLEN et al.) * Claim 1 * ----	3	
			TECHNICAL FIELDS SEARCHED (Int. Cl.5)
			C 08 F 4/00 C 08 F 10/00 C 08 F 110/00 C 08 F 210/00
The present search report has been drawn up for all claims			
Place of search		Date of completion of the search	Examiner
VIENNA		03-12-1991	PUSTERER
CATEGORY OF CITED DOCUMENTS			
X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons * : member of the same patent family, corresponding document			

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